

Clarification the effects of oxygen to carbon ratio on wear mechanism of diamond-like carbon under pressurized hot water

M.Z.M. Rody^{1,*}, K. Okuno¹, N. Umehara¹, N. Inayoshi², K. Sasaki², S. Kawara¹, H. Kousaka¹, X. Deng¹

¹) Department of Mechanical Science and Engineering, Graduate School of Engineering, Nagoya University, Furo-cho, Chikusa-ku, Nagoya, Aichi 464-860, Japan.

²) Materials Engineering R&D Department, DENSO Corporation, 1-1 Showa-cho, Kariya-shi, Aichi 448-8661, Japan.

*Corresponding e-mail: rody@utem.edu.my

Keywords: DLC; wear; oxygen concentration

ABSTRACT – In this paper, the influence of dissolved oxygen concentration amount by means of sputtering depth profile on the wear rate of a hydrogenated Diamond-like Carbon coating (commonly known as DLC) has been studied. The sliding test was performed using a self-developed sliding tester move like pendular, where the DLC coated pins was slid on stainless steel plate (SUS316) at 23°C, 100°C, 200°C, and 300°C in pressurized water environment. The results show that the most crucial factor, affecting the wear rate of DLC coating under sliding in water, divided by two categories. Effect of mechanical wear (graphitization and adhesive wear) was a dominant for low dissolved oxygen concentration. In contrast, above 200°C in rich O₂ water temperature, the graphitization and oxidation wear of DLC has a dominant role on DLC wear rate where friction enhanced oxidation with temperature.

1. INTRODUCTION

Diamond-like Carbon (DLC) has been used as coatings for other materials due to its properties which have high hardness, chemical inertness, low friction and wear rate. There were many more benefits by using the DLC since its properties can be tailored to suit the desired application. Several tribology studies have been done on DLC in various atmosphere and conditions. The specific wear rate of DLC not influenced by hydrogen content in water and air at atmosphere pressure but wear rate tended to decrease with increasing film hardness [1]. Tribological behavior of DLC in hard water is better than soft water environment when protective tribolayer containing Ca and O formed on steel and brass surfaces at elevated temperature 20°C, 50°C and 80°C (0.1MPa) reduces the DLC wear rate and friction [2]. Effects of dissolved oxygen (DO₂) (0.01mg/L, 0.1mg/L, 7.0mg/L) and water pressure (0.1MPa, 1MPa, 19MPa) were also examined briefly but no significant effect was observed. Previous report from both group of researchers shows same argument where temperature was the major role of increment of wear rate at 23°C, 100°C, and 300°C [2,3]. Effect of pressure was small compared to temperature. However, rich content of DO₂ (35-40mg/L) at 300°C increased the specific wear rate of DLC in pure water. DO₂ contents play a significance role on wear of DLC in water. They found that high oxygen elements at topmost layer of DLC increased the wear rate of DLC by using AES

sputtering methods [3]. It is important to clearly understand the DO₂ effect on DLC wear at room temperature up to high temperature in water environment. Thus, the aim of this research is to determine the effect of oxygen to carbon ratio at 100°C, 200°C, 250°C and 300°C for poor O₂ and rich O₂ content at 10MPa water pressure in order to study the wear mechanism of DLC and its counterparts SUS316 by observing the worn surfaces of DLC and its counterpart.

2. EXPERIMENTAL METHOD

The friction test was conducted by using self developed autoclave tribotester. The autoclave body is made from stainless steel (SUS316). The chamber was filled with pure water up to 3/4 of chamber volume or 380ml of water. Two cases of the experiment were studied. One was poor O₂ and the other was rich O₂ content. For poor O₂, the nitrogen gas (N₂) is bubbled into the chamber up to 5MPa through the inlet pipe valve and then it was released in order to minimize the unwanted gases content in the chamber. The outlet valve was closed and 10 MPa of N₂ gas was pressurized into the chamber. The chamber was driven like a pendular by electric motor with 20rpm around 2700 cycles for 200m of sliding distance. In the chamber the DLC pins was slid in a reciprocating motion. The normal force of pins, *N* (load) was 0.056 N. The temperature of the chamber was controlled by heating coils. The test was run at 100°C, 200°C, 250°C and 300°C at 10MPa in water. The procedures were repeated in rich O₂ cases. In rich O₂ cases, the O₂ gases was used as bubbling and pressurizing of water. The wear rate of the DLC film was calculated from the wear diameter measured from the optical microscope by using Archard's formula. In order to evaluate the structure and chemical changes on DLC and SUS316 plate, Raman spectroscopy and EDS SEM analysis were carried out before and after the sliding tests. Auger electron spectroscopy was used in order to examine the tribo layer depth profile for C and O elements at topmost layer of DLC film where Argon gas was for sputtering.

3. RESULTS AND DISCUSSION

The wear rate of DLC under water in rich O₂ and poor O₂ show dependency to temperature as shown in Figure 1. The wear rate was linearly increased with

temperature for poor O₂. In contrast, the wear rate with rich O₂ initiated with the same wear rate at 100°C with poor O₂ and sharply increases above 200°C. Above 200°C with rich O₂ the wear accelerated and reached a maximum of $9.11 \times 10^{-7} \text{mm}^3/\text{Nm}$ of wear rate at 300°C which is 2.2 times of the same temperature with poor O₂ water. According to AES sputtering results in Figure 2 and 3, the ratio of oxygen to carbon O/C for poor O₂ was not having much different for inside wear and outside wear area as compared to rich O₂ cases for all the temperatures. The O/C ratio in DLC wear scar for poor O₂ shows no clear depth dependence in various temperature. On the other hand, for rich O₂, the topmost layer of DLC contained a rich O/C ratio above 200°C water temperature. The maximum O/C ratio for rich O₂ was 0.188 at 300°C. It was 47 times higher than poor O₂ at the same temperature.

It is believed that graphitization occurred during sliding and hard carbon films change to soft associated with temperature [4]. High temperature made the DLC structure softer due to graphitization that allows C bonds weaken, producing cavities on one surface and depressions on the second surface. Adhesion particles remove and rub against the surface, contributing to wear for poor O₂ and rich O₂ water. In rich O₂, the friction enhanced the oxidation which reactively reacts with water temperature led to high wear. The oxidation is controlled by diffusion rate of reactants thus increases exponentially dependent on temperature [5].

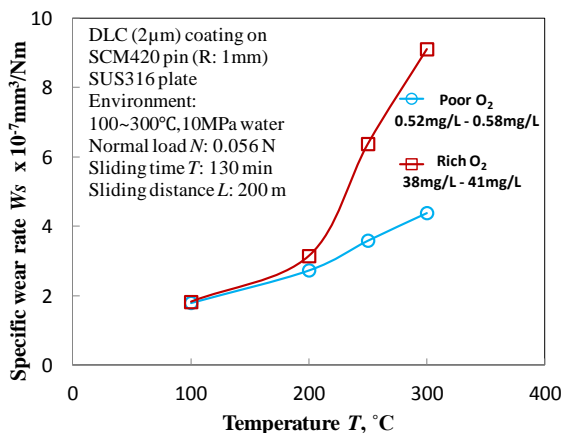


Figure 1 Specific wear rate of DLC in water at various temperature with different oxygen.

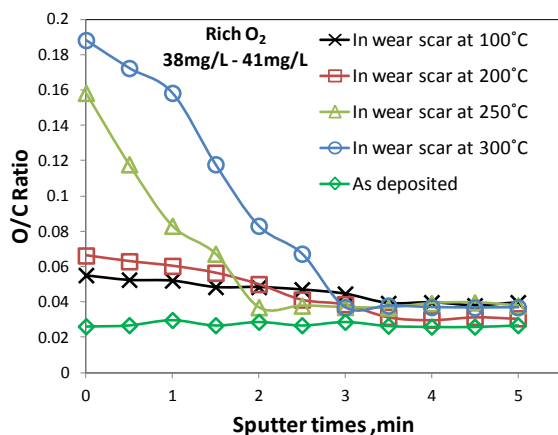


Figure 2 O/C ratio in DLC wear scar for rich dissolved O₂.

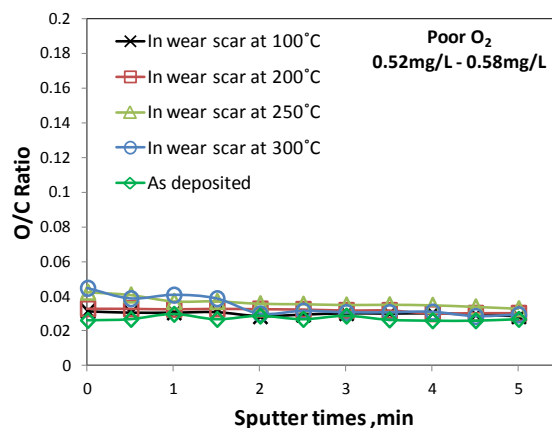


Figure 3 O/C ratio in DLC wear scar for poor dissolved O₂.

4. CONCLUSION

The range of turning parameter obtained from the simulation that can produced dimple structure for A3xx casting Aluminum alloy are cutting speed of 4 – 35 m/min, feed rate of 0.01 – 0.6 mm/rev and depth of cut of 0.01 – 0.05 mm.

5. ACKNOWLEDGEMENT

The author M.Z.M. Rody gratefully acknowledges the scholarship from Ministry of Education Malaysia and Universiti Teknikal Malaysia Melaka for his Doctoral study.

6. REFERENCES

- [1] M. Suzuki, T. Ohana, and a. Tanaka, “Tribological properties of DLC films with different hydrogen contents in water environment,” *Diam. Relat. Mater.*, vol. 13, no. 11–12, pp. 2216–2220, Nov. 2004.
- [2] M. Uchidate, H. Liu, K. Yamamoto, and a. Iwabuchi, “Effects of hard water on tribological properties of DLC rubbed against stainless steel and brass,” *Wear*, vol. 308, no. 1–2, pp. 79–85, Nov. 2013.
- [3] Y. Yagi, H. A. Tasmemir, T. Tokoroyama, N. Umehara, N. Inayoshi, and K. Sasaki, “Clarification of the wear property of DLC under pressurized hot water at 30 MPa and 300°C,” 2013, pp. 2–3.
- [4] S. Zhang, X. L. Bui, and X. Li, “Thermal stability and oxidation properties of magnetron sputtered diamond-like carbon and its nanocomposite coatings,” *Diam. Relat. Mater.*, vol. 15, no. 4–8, pp. 972–976, 2006.
- [5] F. H. Stott, “High-temperature sliding wear of metals,” *Tribol. Int.*, vol. 35, pp. 489–495, 2002.